# Antireflection and SiO<sub>2</sub> Surface Passivation by Liquid-Phase Chemistry for Efficient Black Silicon Solar Cells

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Abstract — We report solar cells with both black Si antireflection and SiO<sub>2</sub> surface passivation provided by inexpensive liquid-phase chemistry, rather than by conventional vacuum-based techniques. Preliminary cell efficiency has reached 16.4%. Nanoporous black Si antireflection on crystalline Si by aqueous etching promises low surface reflection for high photon utilization, together with lower manufacturing cost compared to vacuum-based antireflection coating. Agnanoparticle assisted black Si etching and post-etching chemical treatment recently developed at NREL enables excellent control over the pore diameter and pore separation. Performance of black Si solar cells including open-circuit voltage (Voc) and short-circuit current density  $(J_{SC})$  and blue response has benefited from these improvements. Prior to this study our black Si solar cells were all passivated by thermal SiO<sub>2</sub> produced in tube furnaces. While this passivation is effective, it is not yet ideal for ultra-low-cost manufacturing. In this study we report, for the first time, the integration of black Si with a proprietary liquid phase deposition (LPD) passivation from Natcore Technology. The Natcore LPD forms a layer of 8- to 10-nm SiO<sub>2</sub> on top of the black Si surface in a relatively mild chemical bath at room temperature. We demonstrate black Si solar cells with LPD SiO<sub>2</sub> with a spectrum-weighted average reflection lower than 5%, similar to the more costly thermally-grown SiO<sub>2</sub> approach. However, LPD SiO<sub>2</sub> provides somewhat better surface passivation quality according to the lifetime analysis by the photo-conductivity decay measurement. Black Si solar cells with LPD SiO<sub>2</sub> passivation also demonstrate higher spectral response at short wavelength compared to those passivated by thermally grown SiO<sub>2</sub>. With further optimization, the combination of aqueous black Si etching and LPD could provide a pathway for low-cost, high-efficiency crystalline Si solar cells.

*Index Terms* — black silicon, metal-assisted porous silicon etching, antireflection, liquid phase deposition, surface passivation, photovoltaic cells.

# I. INTRODUCTION

Nanoporous black Si by metal-assisted wet chemical etching has demonstrated the potential of being a viable alternative to conventional vacuum-based antireflection coatings [1, 2]. The nanoporous structure has feature size less than the wavelength of the incident light, and creates a

density-graded surface that suppresses photon reflection by eliminating any abrupt change of index of refraction at the interface [3, 4]. The early development described in Refs. 1 and 2 utilized Au-assisted nanoporous etching to create a black Si surface with pore diameter of ~ 50 nm. The corresponding black Si solar cells have averaged surface reflection lower than 3% but poor spectral response at short wavelength (blue response) which limits the short-circuit current density ( $J_{SC}$ ). High surface area from the nanoporous structure is one of the major factors that attributes to the poor blue response.

In order to better control the nanoporous feature size and reduce the cost of the catalytic etching we recently developed Ag-assisted black Si etching and subsequent chemical surface treatment at NREL [5]. The pore diameter can be 50 to 100 nm or beyond, and the blue response improves accordingly.

High quality surface passivation is an essential for good blue response. A thin layer of 10 to 20-nm thermally grown SiO<sub>2</sub> has been used at NREL to passivate the front surface of black Si solar cells. Thermally grown SiO<sub>2</sub> forms a conformal coverage on Si nanostructure [6] and provides decent passivation quality [1, 2]. However, this step often takes a separated high-temperature oxidation and could erode the cost benefit of the liquid-phase black Si antireflection. A proprietary liquid phase deposition (LPD) currently being commercialized by Natcore Technology, on the other hand, deposits a layer of SiO<sub>2</sub> on candidate substrate in reactive solution and it represents a promising low-cost route to manufacturing of black Si photovoltaic. Here we report promising characteristics of the LPD-SiO<sub>2</sub>-passivated black Si solar cells including their surface reflectance, photoconductivity decay (PCD) lifetime, spectral response and solar cell performance.

## **II. EXPERIMENT**

The Si substrate is double-sided polished boron-doped ptype Si (100) FZ with resistivity of ~ 2.8  $\Omega$ -cm and thickness of ~ 300  $\mu$ m. The backside of the Si is protected so that Agassisted orous black silicon etching creates black Si only on the front side. The etching consists of two steps; firstly the Ag nanoparticles are electrolessly deposition on Si substrate from AgNO<sub>3</sub> and HF solution, and secondly preferential nanoporous etching is performed by immersing the Agnanoparticle decorated Si in diluted HF and  $H_2O_2$  mixture. The duration of Ag deposition and the subsequent porous Si etching dictate the diameter, separation and the depth of the nanopores. This two-step Ag-assisted etching creates porous Si with feature size well below the wavelength of incident light that a Si photovoltaic cell is able to utilize. The random pore depth results in a near linear density-graded Si surface that suppresses the reflection.

As mentioned previously it is crucial to minimize the surface recombination to obtain high conversion efficiency. Here we intentionally reduce the surface area of the pore structure by immersing the black Si sample in TMAH solution that etches Si and widens the pores. Two different TMAH treatments, Treatment I & II, are performed. The primary purpose of the treatment is to reduce the surface area of the porous black Si. Treatment I and II differ on the duration of the TMAH etching and when it is performed among the solar cell fabrication.

An n-type emitter is formed by POCl<sub>3</sub> diffusion at 850°C with sheet resistance of ~ 80  $\Omega/\Box$ . After stripping the PSG layer in dilute HF the surface of black Si samples are either passivated by LPD or thermal SiO<sub>2</sub>. LPD is performed in reactive solution containing H<sub>2</sub>SiF<sub>6</sub>, SiO<sub>2</sub> powder and H<sub>2</sub>O. As the reaction progresses SiO<sub>2</sub> deposits on the black Si surface and we control the thickness of the LPD SiO<sub>2</sub> to be less than 10 nm in this study. At this moment we have found that the LPD-SiO<sub>2</sub> needs to be annealed in an oxygen-containing ambient followed by forming gas annealing in order to achieve the best surface passivation quality. The control samples, on the other hand, have the black Si passivated by a thicker thermal SiO<sub>2</sub> (25-30 nm) separately grown at 850°C in dry O<sub>2</sub> ambient.

A full-area Al-BSF and lithographically patterned metal grid finishes the back and front contact of the solar cells, respectively. The cell area is  $\sim 1 \text{ cm}^2$ . There is no additional antireflection coating on either type of nanostructured solar cell under study.

The reflectance is measured by Varian Cary 6000i spectrophotometer with an integrating sphere, and the PCD-lifetime is measured by Sinton WCT-120 lifetime tester. Current density-voltage (J-V) characteristic is measured by a calibrated XT-10 1-sun solar simulator.



Fig. 1: Reflectance of black Si solar cells passivated by LPD and thermally grown  $SiO_2$ .

#### III. RESULT

Figure 1 shows the reflectance of black Si solar cells, LPD-SiO<sub>2</sub> passivated and control (thermal SiO<sub>2</sub>), with TMAH Treatment II. . Both the thermal SiO<sub>2</sub> control and the LPD sample have similarly low reflectance indicating that the LPD does not interfere with the antireflection of the black Si surface. Minor difference in the reflectivity spectrum is likely the result of poor uniformity control during black Si etching. Spectrum-weighted average reflectance from 350- to 1000-nm wavelength is 5% and 4% for LPD and thermal SiO<sub>2</sub> black Si, respectively. The LPD black Si solar cell with Treatment I has spectrum-weighted average reflectance of 4%, again similar to the control. Figure 2 presents the implied  $V_{OC}$  obtained from PCD-lifetime measurement of the passivated one-sided black Si samples with n-type emitter on both sides. LPD SiO<sub>2</sub> demonstrates somewhat better passivation quality than the thermal SiO<sub>2</sub> in both treatments.

Emitter dark saturation current density  $(J_{oe})$  extracted from the PCD-lifetime measurement is 370 and 180 fA/cm<sup>2</sup> for LPD black Si samples that have undergone Treatments I and II, respectively. They are about 5 times higher than the reported  $J_{oe}$  on SiO<sub>2</sub>-passivated POCl<sub>3</sub>-diffused Si surface with similar emitter sheet resistance [7]. The higher  $J_{oe}$  reflects the intrinsic surface passivation quality as well as the inevitable impact of the large surface area of nanoporous black Si. Nevertheless, we believe that a further optimization of black Si etching, TMAH treatment, and LPD passivation can provide both excellent antireflection and solar cell performance.



Fig. 2: Implied V<sub>OC</sub> of passivated one-sided black Si samples.



Fig. 3: J-V curves of LPD black Si solar cells.

Figure 3 presents J-V curves of LPD black Si solar cells measured under simulated 1-sun conditions at 25°C. LPD black Si solar cell with Treatment II has higher energy conversion efficiency than the solar cell with Treatment I, mainly due to the better surface passivation quality seen in Fig. 2. The best LPD black Si solar cell (Treatment II) has  $V_{OC}$  of 607.4 mV,  $J_{SC}$  of 34.9 mA/cm<sup>2</sup>, FF of 77.2%, and efficiency of 16.4%. Its  $J_{SC}$  and FF are comparable to the thermal SiO<sub>2</sub> black Si counterpart. However, the  $V_{OC}$  of the LPD cell is about 20 mV below the thermal SiO<sub>2</sub> control, even though the implied  $V_{OC}$  is a bit higher than the thermal SiO<sub>2</sub> (Fig. 2).

Intrinsic quantum efficiency (IQE) shown in Fig. 4 compares spectral response of LPD- and thermal-SiO<sub>2</sub> black Si solar cells with Treatment I and II, respectively. The IQE fluctuation at near IR is still under investigation but most likely related to the Al-BSF formation. We believe it is the poor near IR response that contributes to low  $V_{OC}$  of the LPD black Si solar cell by Treatment II as mentioned previously. Nonetheless, the superior blue response of the LPD black Si

solar cell in both cases indicates better passivation quality by LPD SiO<sub>2</sub>.



Fig. 4: Intrinsic quantum efficiency of LPD and thermal  $SiO_2$  black Si solar cells treated by (A) Treatment I, and (B) Treatment II.

# **IV. CONCLUSION**

Nanoporous black Si solar cells integrated with low-cost, solution-based LPD SiO<sub>2</sub> passivation is reported for the first time. The combination demonstrates promising results; the LPD solar cells (1) retain low reflectivity of the black Si surface, and (2) provide better surface passivation quality compared with the thermally grown SiO<sub>2</sub> as seen by PCD-lifetime and spectral response measurements. Further optimization of the key processing steps such as black Si formation, LPD, and TMAH treatment could promise an extremely low-cost and high-efficiency black Si solar cell.

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## REFERENCES

[1] H.-C. Yuan, V.E. Yost, M.R. Page, P. Stradins, D.L. Meier, and H.M. Branz, "Efficiency black silicon solar cell with a density-graded nanoporous surface: optical properties, performance limitations, and design rules", *Appl. Phys. Lett.*, vol. 95, pp. 123501, 2009,.

[2] F. Toor, H.M. Branz, M.R. Page, K.M. Jones, and H.-C. Yuan, "Multi-scale surface texture to improve blue response of nanoporous black silicon solar cells", *Appl. Phys. Lett.*, vol. 99, pp. 103501, 2011.

[3] R.B. Stephens, and G.D. Cody, "Optical reflectance and transmission of a textured surface", *Thin Solid Films*, vol. 45, pp. 19-29, 1977.

[4] H.M. Branz, V.E. Yost, S. Ward, K.M. Jones, B. To, and P. Stradins, "Nanostructured black silicon and the optical reflectance of graded-density surfaces", *Appl. Phys. Lett.*, vol. 94, pp. 231121, 2009.

[5] J. Oh, F. Toor, H.-C. Yuan, and H.M. Branz, "High – efficiency black silicon solar cells with no antireflection coating", 21<sup>st</sup> Workshop on Crystalline Silicon Solar Cells & Modules: Materials and Processes

[6] Y. Yan, H.-C. Yuan, V.E. Yost, K. Jones, M. Al-Jassim, an dH.M. Branz, "Microstructure and surface chemistry of nanoporous 'black silicon' for photovoltaics," *35<sup>th</sup> IEEE Photovoltaic Specialist Conference*, 2010, p. 002255.

[7] A. Cuevas, P.A. Basore, G. Giroult-Matlakowski, and C. Dubois, "Surface recombination velocity of highly doped n-type silicon", *J. Appl. Phys.*, vol. 80, pp. 3370-3375, 1996.